

Syntheses, Crystal and Electronic Structures, and Linear Optics of LiMBO₃ (M = Sr, Ba) Orthoborates

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The syntheses, crystal and electronic structures, and linear optical properties of the new orthoborates LiMBO₃ (M = Sr, Ba) are reported here. These compounds, which crystallize in the monoclinic space group $P2_1/n$ with cell dimensions $a = 6.476(2)$, $b = 6.684(3)$, $c = 6.843(3)$ Å, $\beta = 109.41(3)^\circ$, and $Z = 4$ for M = Sr, $a = 6.372(1)$, $b = 7.022(3)$, $c = 7.058(1)$ Å, $\beta = 113.89(1)^\circ$, and $Z = 4$ for M = Ba, are modeled in terms of the cluster units (LiMBO₃)₂. The calculated electronic structures show that the top of the valence band consists of mostly the O 2p orbitals and the bottom of the conduction band consists of cationic orbitals. The dynamic refractive indices of these orthoborates are obtained in the framework of the INDO/SCI approximation together with the “sum-over-states” method. It is found that the refractive index is larger for LiSrBO₃ than for LiBaBO₃ and the charge transfer from O²⁻ anionic orbitals to cationic orbitals appears to provide significant contribution to the linear polarizability of these compounds.

1. Introduction

Research efforts have been directed at solid-state borates due to a variety of physical and chemical features exhibited by these compounds, such as luminescence of doped borates SrB₄O₇^{1,2} and X₂Z(BO₃)₂ (X = Ba, Sr, Ca; Z = Ca, Mg),³ nonlinear optical properties⁴ of borate compounds including the (B₃O₆)³⁻ group, and catalytic activity of the zinc–copper borate ZnCu₂(BO₃)₂.⁵ Among nonlinear optical crystals, solid-state borates have emerged as the pre-eminent materials for high-power applications, and now, they are being utilized in the manufacture of components in essentially all complex microprocessor-based devices from PCs to cellular phones. An understanding of the relationship between microscopic structure and macroscopic properties has been recognized as an important aid in the design and improvement of materials.⁶ Cheng et al. have calculated the electronic structures and nonlinear optical coefficients of MB₆O₁₀ (M = Cs₂, Li₂, CsLi) with an aim to understand the electronic origin of their optical susceptibility.⁷ Keszler and co-workers have systematically investigated the syntheses, structures, and properties of alkali-metal borates searching for a trend in the structure–property relationship.^{8,9}

In this study, we first report the syntheses and single-crystal structural determinations for mixed alkaline and alkaline-earth metal orthoborates LiMBO₃ (M = Sr, Ba), and we find that the structural types of our obtained compounds are different from those of the LiABO₃ orthoborates (A = Mg, Mn, Co, Zn, and Cd) reported in references.^{10–13} Belkebir and co-workers have calculated the crystallographic cell of the phases of LiZnBO₃ obtained by solid-state reaction without melting by indexing their X-ray power diffraction patterns, and the two structures found are monoclinic probably with the same Li–Zn cationic disorder as evidenced by vibrational behavior.¹³ Piffard and co-workers have determined the structure of LiCoBO₃ to be of the space group $C2/c$ and indicated that the Co and Zn cations have occupied similar positions within the trigonal bipyramids above and below the center plane in LiCoBO₃ and LiZnBO₃, respectively.¹⁰ Then, we use a cluster unit representing the crystalline orthoborate lattice to calculate the electronic structure and refractive indices of mixed alkaline-earth metal and lithium orthoborates LiMBO₃ (M = Sr, Ba). In this way, we obtain a means to examine structural and compositional contributions to linear optical properties. The calculated results show a larger refractive index for LiSrBO₃ than for LiBaBO₃ and indicate that the charge transfers from O 2p orbitals to cation valence orbitals have major contributions to the refractive indices of LiMBO₃ (M = Sr, Ba).

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(1) Blasse, G.; Dirksen, G. J.; Meijerink, A. *Chem. Phys. Lett.* **1990**, *167*, 41–44.

(2) Meijerink, A.; Nuyten, J.; Blasse, G. *J. Lumin.* **1989**, *44*, 19–31.

(3) Versteegen, J. M. P. J. *J. Electrochem. Soc. Solid-State Sci. Technol.* **1974**, *121*, 1631–1633.

(4) Chen, C.-Z.; Gao, D.-S.; Chen, C.-T. *Acad. Thesis Conf. Cryst. Growth Mater. (China)* **1979**, *B44*, 107–111.

(5) Zletz, A. U.S. Patent Application, 709, 790, March 11, 1985, Amoco Corp.

(6) Munowitz, M.; Jarman, R. H.; Harrison, J. F. *Chem. Mater.* **1993**, *5*, 661–671; **1993**, *5*, 1257–1267.

(7) Cheng, W.-D.; Chen, J.-T.; Lin, Q.-S.; Zhang, Q.-E.; Lu, J.-X. *Phys. Rev. B* **1999**, *60*, 11747–11754.

(8) Akella, A.; Keszler, D. A. *J. Solid State Chem.* **1995**, *120*, 74–79.

(9) Smith, R. W.; Keszler, D. A. *J. Solid State Chem.* **1997**, *129*, 184–188.

(10) Piffard, Y.; Rangan, K. K.; An, Y.; Guyomard, D.; Tournoux, M. *Acta Crystallogr.* **1998**, *C54*, 1561–1563.

(11) Norrestam, R. *Z. Kristallogr.* **1989**, *187*, 103–110.

(12) Sokolova, E. V.; Simonov, M. A.; Belov, N. V. *Z. Kristallogr.* **1980**, *25*, 1285–1286.

(13) Belkebir, A.; Tarte, P.; Rulmont, A.; Gilbert, B. *New J. Chem.* **1996**, *20*, 311–316.

Table 1. Crystal Parameters

formula	LiSrBO ₃	LiBaBO ₃
fw	153.37	203.09
space group	<i>P2</i> (1)/ <i>n</i>	<i>P2</i> (1)/ <i>n</i>
<i>a</i> (Å)	6.4800(13)	6.372(1)
<i>b</i> (Å)	6.6800(13)	7.022(3)
<i>c</i> (Å)	6.8400(14)	7.058(1)
β (deg)	109.41(3)	113.89(1)
<i>V</i> (Å ³)	279.25(10)	288.7(2)
<i>Z</i>	4	4
<i>D</i> _{calc} (mg/m ³)	3.648	4.67
<i>R</i>	0.0443	0.0551
<i>R</i> _w	0.0898	0.0626
<i>S</i>	0.982	0.93

2. Experimental and Computational Procedures

2.1. Syntheses and Crystal Growths. **2.1.1. LiSrBO₃.** A mixture containing appropriate amounts of LiCO₃ (Chemical pure), SrCO₃ (Chemical pure), and H₃BO₃ (Analytical reagent) was ground into fine powder in a mortar of agate. This mixture was heated to 450 °C in a platinum crucible and kept at this temperature for 4 h, followed by heating at 840 °C for 10 h and at 910 °C for 24 h. The mixture, then, was cooled from 910 to 600 °C at a rate of 2.7 °C h⁻¹. It finally was quenched to room temperature. A few colorless pillar crystals were found from the melt of the mixture.

2.1.2. LiBaBO₃. A stoichiometric mixture of LiCO₃ (0.74 g, chemical pure), BaF₂ (1.71 g, analytical reagent), SrCO₃ (1.48 g, chemical pure), and H₃BO₃ (1.24 g, analytical reagent) was ground into fine powder in a mortar of agate, then heated to 450 °C in a platinum crucible, and kept at this temperature for 4 h, followed by heating at 840 °C for 24 h and at 910 °C for 24 h, then cooled from 910 to 800 °C at a rate of 1.0 °C h⁻¹ and from 800 to 600 °C at a rate of 4.0 °C h⁻¹, and finally air-quenched to room temperature. A few colorless pillar crystals were found from the melt.

2.2. X-ray Determination. A single crystal of LiSrBO₃ and LiBaBO₃ with approximate dimensions 0.18 × 0.08 × 0.06 and 0.30 × 0.20 × 0.10 mm³ was selected for single-crystal X-ray diffraction, respectively. The diffraction data were collected on an Enraf-Nonius CAD4 diffractometer with graphite monochromator Mo K α radiation for these two crystals. Cell constants were obtained from least-squares refinement, using the setting angles of 25 reflections in the range 26° < 2 θ < 48° for LiSrBO₃ and 25 reflections in the range 22° < 2 θ < 46° for LiBaBO₃. The crystallographic parameters of these two crystals are listed in Table 1. The intensity data were collected at 273 K in the range -11 ≤ *h* ≤ 11, 0 ≤ *k* ≤ 12, and 0 ≤ *l* ≤ 12 for LiSrBO₃ and the range 0 ≤ *h* ≤ 12, 0 ≤ *k* ≤ 13, and -13 ≤ *l* ≤ 13 for LiBaBO₃, using the $\omega/2\theta$ scan technique with a scan speed of 5°/min and a scan width of $\Delta\omega = (0.8 + 0.35 \tan \theta)^\circ$, respectively. The intensities of three standard reflections were measured every 60 min, and the intensity decay was 0.6 and 4.7% for crystals LiSrBO₃ and LiBaBO₃, respectively. Lorentz and polarization corrections were applied to the data. The linear absorption coefficients are 14.3 and 135.3 mm⁻¹ for these two crystals, respectively. An empirical absorption correction based on a ψ -scan was applied and the relative transmission coefficients ranged from 0.324 to 1.00 with an average value of 0.662 and from 0.484 to 0.996 with an average value of 0.740, respectively. The 1824 and 2301 reflections were used to measured with 2 θ_{\max} = 80°; 1223 and 1490 reflections with *I* > 3 σ (*I*) were used in structural determination and refinement for LiSrBO₃ and LiBaBO₃ crystals, respectively.

The structures of LiSrBO₃ and LiBaBO₃ were separately determined by the Shelx/PC and MolEN/PC programs. From the systematic absence of *h0l*: *l* = 2*n*; *0k0*: *k* = 2*n* and from subsequent least-squares refinement, the space group of these two crystals was determined to be *P2*₁/*n*. Note that this space group is not a standard space group. However, it is often convenient and the angle of β will not become much larger than 90° when we prefer the space group *P2*₁/*n* in determining the crystal structures. The Sr and Ba atoms were located from

Table 2. Fractional Atomic Coordinates and Equivalent Isotropic Displacement Parameters (Å²)

atom	<i>x</i>	<i>y</i>	<i>Z</i>	<i>U</i> _{eq} ^a
LiSrBO ₃				
Sr	0.20197(5)	0.12378(5)	0.86760(5)	0.00811(9)
O(1)	0.1560(5)	0.1105(4)	1.2178(4)	0.0130(5)
O(2)	0.1073(5)	0.2993(4)	0.4940(4)	0.0109(4)
O(3)	0.1963(5)	0.4897(4)	0.9367(4)	0.0109(5)
B	0.1886(6)	0.1326(6)	0.4270(6)	0.0084(5)
Li	0.1002(13)	0.4086(10)	1.1930(11)	0.0136(13)
LiBaBO ₃				
Ba	0.33502(8)	0.13980(7)	0.15907(6)	0.01093(6)
O(1)	0.3793(9)	0.3180(7)	0.5164(8)	0.0085(9)
O(2)	0.291(1)	0.1598(9)	0.7740(8)	0.017(1)
O(3)	0.2199(9)	0.0011(7)	0.4569(7)	0.0074(9)
B	0.296(1)	0.160(1)	0.579(1)	0.007(1)
Li	0.896(2)	0.071(2)	0.298(2)	0.009(3)

$$^a U_{eq} = 1/3 \sum_i \sum_j U^{ij} \alpha_i^* \alpha_j^* a_i a_j$$

the direct method; the remaining atoms were located in succeeding difference Fourier synthesis. The final full-matrix least-squares refinement for 56 and 55 variable parameters converged to *R* = 4.43%, *R*_w = 8.98% [in which $w = 1/(\sigma^2(F_o^2) + (aP)^2 + bP)$; $P = (2F_c^2 + \text{Max}(F_o^2, 0))/3$], *S* = 0.982, and (Δ/σ)_{max} = 0.0001 for LiSrBO₃ and converged to *R* = 5.51%, *R*_w = 6.26% [in which $w = 1/(\sigma^2(F) + (0.020F)^2 + 1.0)$], *S* = 0.93, and (Δ/σ)_{max} = 0.0001 for LiBaBO₃. Neutral atomic scattering factors were taken from Cromer and Waber.¹⁴ The maximum and minimum peaks on the final different Fourier map are 1.43 and -1.00 e/Å³ and 5.57 and -1.72 e/Å³ for SrLiBO₃ and BaLiBO₃ crystals, respectively. The atom coordinates and thermal parameters are listed in Table 2. Here, it is noted that the BaF₂ attempted as a flux was added into the mixture of SrCO₃, LiCO₃, and H₃BO₃, and the variation of temperature was controlled to try and obtain a new phase of LiSrBO₃. However, the crystal of LiBaBO₃ was found in the X-ray structural determination and the atom of Sr was not present in this crystal. We made a trial structure including the Sr atoms in the LiBaBO₃ during the solution of crystal structure by the Shelxtl/PC program. The result showed a large value of *R* and unreasonable coordinations of metal atoms. Accordingly, we believe no Sr atoms are in this crystal structure.

2.3. Computational Details. The wave functions and energies obtained from electronic structural calculations were employed to compute the polarizabilities of clusters, and the electronic structural calculations of the clusters were based on an all-valence-electron, semiempirical INDO self-consistent field (SCF) molecular orbital (MO) procedure with configuration interaction (CI) modified by Zerner and co-workers.¹⁵⁻¹⁸ There are the one-center core integral *U*_{μμ}, resonance integral $\beta_{\mu\nu}$, two-electron integral $\gamma_{\mu\nu}$, overlap integral *S*_{μν}, and density matrix element **P**_{μν} in the matrix element of the Fock operator under the INDO approximation. The INDO model as employed herein included all one-center two-electron integrals and two-center two-electron integrals $\gamma_{\mu\nu}$. The one-center two-electron integrals $\gamma_{\mu\mu}$ were chosen from the Pariser approximation, $\gamma_{\mu\mu} = F^0(\mu\mu) = IP_{\mu} - EA_{\mu}$, and the two-center two electron integrals were calculated using the Mataga-Nishimoto formula, $\gamma_{\mu\nu} = 1.2/[R_{AB} + 2.4/(\gamma_{\mu\mu} + \gamma_{\nu\nu})]$ in the spectroscopic version of the INDO method. The Slater orbital exponents ζ and the other calculating parameters are listed in Table 3. The molecular orbital calculations were performed by the restricted Hartree-Fock method. The ground state was constructed as a single

(14) Cromer, D. T.; Waber, J. T. In *International Table for X-ray Crystallography*; Kynoch Press: Birmingham, 1974; Vol. IV, Table 2.2A, p 71.

(15) Bacon, A. D.; Zerner, M. C. *Theor. Chim. Acta* **1979**, *53*, 21-54.

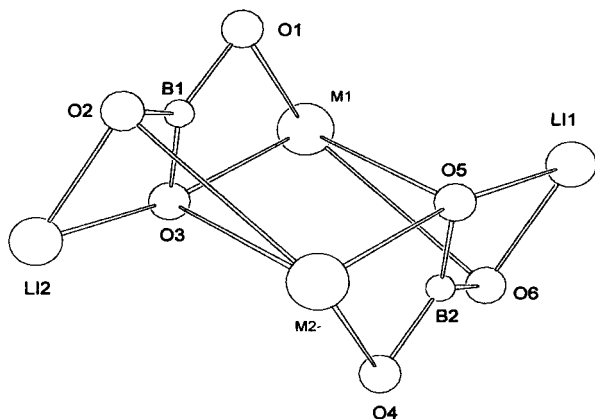
(16) Zerner, M. C.; Lovw, G. H.; Kirchner, R. F.; Mueller-Westerhoff, U. T. *J. Am. Chem. Soc.* **1980**, *102*, 589-599.

(17) Anderson, W. P.; Edwards, E. D.; Zerner, M. C. *Inorg. Chem.* **1986**, *25*, 2728-2732.

(18) Anderson, W. P.; Cundari, T. R.; Zerner, M. C. *Int. J. Quantum Chem.* **1991**, *39*, 31-45.

Table 3. INDO/S Model Parameters

parameter	B	O	Li	Sr	Ba
$\zeta_{rs,mp}$ (\AA^{-1})	1.300	2.275	0.650	1.214	1.263
$\zeta_{(n-1)d}$ (\AA^{-1})				2.058	2.658
$-I_{rs}$ (eV)	14.05	32.90	5.41	5.84	5.21
$-I_{rp}$ (eV)	8.70	17.28	3.61	3.76	3.43
$-I_{(n-1)d}$ (eV)				3.66	3.22
$-\beta_{rs,mp}$ (eV)	17.00	34.20	9.00	1.88	2.66
$-\beta_{(n-1)d}$ (eV)				10.05	12.50
$\gamma_{rs,mp}$ (eV)	8.68	13.00	4.57	3.75	4.68
$\gamma_{(n-1)dd}$ (eV)				5.31	5.19
$\gamma_{(n-1)drs,mp}$				4.53	3.79

Figure 1. Selected cluster unit model of (LiMBO_3) .

determinant from the Hartree–Fock SCF calculated results. Only single-substituted determinants relative to the ground state configuration were considered and only singlet spin-adapted configurations needed to be included in the CI calculations. The ground state and all excited states had the multiplicity of 1. The electron was promoted from the 13 highest occupied orbitals to the 13 lowest unoccupied orbitals and the configuration space was constructed by these 26 active orbitals. The wave functions and energy eigenvalues of the excited states were determined by solving the secular equation relating to configuration coefficients. The dipole and transition moment matrix elements were expressed as a sum of one-electron integrals.

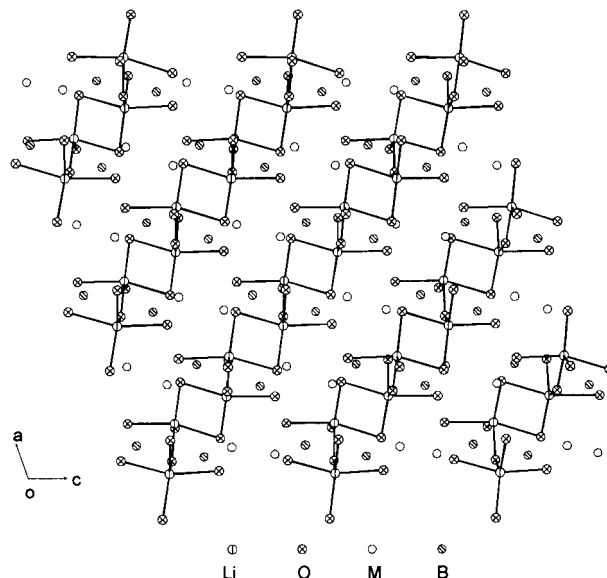
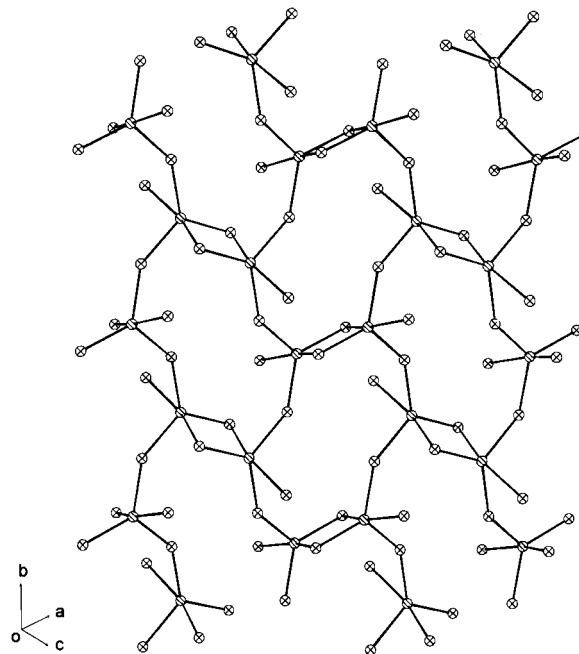
The cluster units $(\text{LiSrBO}_3)_2$ and $(\text{LiBaBO}_3)_2$ representing extended oxide crystals LiSrBO_3 and LiBaBO_3 were selected for calculations, as shown in Figure 1. The coordinate systems are defined in the calculations. The coordinate center is located at the center of the ring $[\text{M1M2O3O5}]$. The x axis is defined to be parallel to the connecting line between the O3 and O5 atoms, and the y axis is laid down the plane constructed by three atoms, O3–O5–M1 in Figure 1. The z axis is defined as the right-hand rule of the coordinate system for these two clusters, respectively. Theoretical calculations of electronic structures and polarizabilities are based on the crystallographic structural data for these two clusters.

The tensor components of the polarizability $\alpha(\omega)$ with frequency dependence for the clusters $(\text{LiSrBO}_3)_2$ and $(\text{LiBaBO}_3)_2$ were calculated by the sum-over-states (SOS) method as follows:

$$\alpha_{ij}(\omega) = 1/\hbar \sum_m \mu_{gm}^i \mu_{mg}^j [(\omega_{mg} - \omega_p)^{-1} + (\omega_{mg} + \omega_p)^{-1}] \quad (1)$$

3. Results and Discussions

3.1. Crystal Structures. Drawings of the contents of the unit cells of compounds LiMBO_3 ($M = \text{Sr}, \text{Ba}$) are shown in Figure 2. It is found that LiMBO_3 structures can be constructed from a stack of $[\text{MO}]$ and $[\text{LiO}]$ layers along the $[101]$ direction and B atoms localized in adjacent layers as a bridging role. In the $[\text{LiO}]$ layers, the adjacent polyhedrons constructed from

Figure 2. Crystal structure stacked from the $[\text{MO}]$ and $[101]$ direction in $[\text{LiMBO}_3]$. The M–O bonds are omitted for clarity.Figure 3. Structure of the $[\text{LiO}]$ layer along the ac diagonal plane in $[\text{LiMBO}_3]$.

the LiO_5 form dimers by sharing edges, and each dimer connects with the four adjacent dimers by the four O atoms to form two-dimensional sheets along the ac diagonal plane, as shown in Figure 3. The Li atoms are coordinated by five O atoms and the LiO_5 polyhedron is a distorted trigonal bipyramid. For the LiSrBO_3 crystal, the Li–O distances vary from 1.955(7) to 2.169(8) \AA with an average value of 2.056 \AA , where the long bond lengths of Li–O are in the axial direction of the trigonal bipyramid. An average bond length of the Li–O in the LiBaBO_3 crystal is close to that of the LiSrBO_3 one. The B–O distances vary from 1.369(4) to 1.385(6) \AA with an average value of 1.377 \AA and the O–B–O angles are between 118.3(7) $^\circ$ and 122.6(7) $^\circ$. These values are normal in a $[\text{BO}_3]$ plane of LiMBO_3 ($M = \text{Sr}, \text{Ba}$). Table 4 lists the interatomic distances and angles of LiMBO_3 . There are different coordinations

Table 4. Interatomic Distances (Å) and Bond Angles (deg)

Interatomic Distances (Å)		Bond Angles (deg)		Interatomic Distances (Å)		Bond Angles (deg)	
a. LiSrBO₃				b. LiBaBO₃			
Sr–O(1)	2.6928(7)	O(1)–Sr–O(1)	79.57(2)	Ba–O(1)	2.726(6)	O(1)–Ba–O(1)	131.2(2)
Sr–O(1)	2.513(1)	O(1)–Sr–O(2)	98.01(2)	Ba–O(1)	2.670(6)	O(1)–Ba–O(1)	89.3(2)
Sr–O(2)	2.689(1)	O(1)–Sr–O(2)	82.02(3)	Ba–O(1)	2.677(5)	O(1)–Ba–O(2)	149.6(2)
Sr–O(2)	2.5171(9)	O(1)–Sr–O(2)	155.856(9)	Ba–O(2)	2.622(6)	O(1)–Ba–O(2)	112.9(1)
Sr–O(2)	2.5297(8)	O(1)–Sr–O(3)	123.03(2)	Ba–O(2)	3.067(8)	O(1)–Ba–O(2)	79.1(2)
Sr–O(3)	2.495(1)	O(1)–Sr–O(3)	93.19(2)	Ba–O(2)	3.030(8)	O(1)–Ba–O(3)	53.0(2)
Sr–O(3)	2.5469(9)	O(1)–Sr–O(2)	149.26(1)	Ba–O(3)	2.675(6)	O(1)–Ba–O(3)	78.3(2)
Li–O(1)	2.0219(9)	O(1)–Sr–O(2)	79.05(1)	Ba–O(3)	2.646(5)	O(1)–Ba–O(3)	68.4(2)
Li–O(1)	2.0131(5)	O(1)–Sr–O(2)	96.83(2)	Ba–O(3)	3.185(5)	O(1)–Ba–O(1)	69.3(2)
Li–O(2)	2.1664(9)	O(1)–Sr–O(3)	80.857(8)	B–O(1)	1.38(1)	O(1)–Ba–O(2)	75.2(2)
Li–O(3)	2.1250(7)	O(1)–Sr–O(3)	155.81(1)	B–O(2)	1.39(1)	O(1)–Ba–O(2)	73.2(2)
Li–O(3)	1.9544(6)	O(2)–Sr–O(2)	131.318(9)	B–O(3)	1.373(8)	O(1)–Ba–O(2)	140.5(2)
B–O(1)	1.3850(6)	O(2)–Sr–O(2)	96.91(2)	Li–O(1)	2.10(2)	O(1)–Ba–O(3)	78.8(2)
B–O(2)	1.3737(4)	O(2)–Sr–O(3)	74.89(2)	Li–O(2)	1.95(2)	O(1)–Ba–O(3)	133.6(1)
B–O(3)	1.3685(4)	O(2)–Sr–O(3)	54.24(3)	Li–O(2)	1.99(2)	O(1)–Ba–O(3)	104.1(2)
		O(2)–Sr–O(2)	73.87(2)	Li–O(3)	1.97(1)	O(1)–Ba–O(2)	87.8(2)
		O(2)–Sr–O(3)	144.00(2)	Li–O(3)	2.20(2)	O(1)–Ba–O(2)	142.2(2)
		O(2)–Sr–O(3)	77.09(2)			O(1)–Ba–O(2)	145.7(2)
		O(2)–Sr–O(3)	79.276(8)			O(1)–Ba–O(3)	74.6(2)
		O(2)–Sr–O(3)	80.34(2)			O(1)–Ba–O(3)	77.9(2)
		O(3)–Sr–O(3)	121.599(5)			O(1)–Ba–O(3)	144.2(2)
		O(1)–Li–O(1)	122.33(2)			O(2)–Ba–O(2)	86.9(2)
		O(1)–Li–O(2)	69.31(1)			O(2)–Ba–O(2)	86.5(2)
		O(1)–Li–O(3)	103.092(9)			O(2)–Ba–O(3)	152.5(2)
		O(1)–Li–O(3)	120.218(5)			O(2)–Ba–O(3)	71.5(2)
		O(1)–Li–O(2)	99.92(2)			O(2)–Ba–O(3)	125.7(2)
		O(1)–Li–O(3)	69.94(2)			O(2)–Ba–O(2)	71.1(2)
		O(1)–Li–O(3)	117.29(3)			O(2)–Ba–O(3)	94.1(2)
		O(2)–Li–O(3)	162.187(8)			O(2)–Ba–O(3)	134.7(2)
		O(2)–Li–O(3)	105.13(2)			O(2)–Ba–O(3)	44.6(2)
		O(3)–Li–O(3)	92.63(2)			O(2)–Ba–O(3)	119.8(1)
		O(1)–B–O(2)	119.62(1)			O(2)–Ba–O(3)	68.3(2)
		O(1)–B–O(3)	119.07(2)			O(2)–Ba–O(3)	59.3(1)
		O(2)–B–O(3)	121.31(3)			O(3)–Ba–O(3)	123.3(2)
						O(3)–Ba–O(3)	69.6(2)
						O(3)–Ba–O(3)	121.5(2)
						O(1)–B–O(2)	119.0(6)
						O(1)–B–O(3)	122.6(7)
						O(2)–B–O(3)	118.3(7)
						O(1)–Li–O(2)	105.8(6)
						O(1)–Li–O(2)	71.3(5)
						O(1)–Li–O(3)	105.8(8)
						O(1)–Li–O(3)	158.0(7)
						O(2)–Li–O(2)	128.1(8)
						O(2)–Li–O(3)	109.3(7)
						O(2)–Li–O(3)	69.4(6)
						O(2)–Li–O(3)	121.5(7)
						O(2)–Li–O(3)	94.5(7)
						O(3)–Li–O(3)	95.9(6)

between the Sr and Ba atoms in LiMBO₃ (M = Sr, Ba) crystals. The Sr atoms are coordinated by seven O atoms, and the SrO₇ polyhedron may be described as a mono-capped distorted trigonal prism. The Sr–O distances vary from 2.495(3) to 2.692(3) Å with an average value of 2.594 Å, which compared very well to the expected value 2.590 Å calculated from the crystal radii for the seven-coordinate Sr²⁺ ion.¹⁹ In the [SrO] layers, the Sr atoms through sharing O–O edges extend along the *b* direction to form chains, and the adjacent chains link together by O atoms to form puckering sheets parallel to the *ac* diagonal plane, as shown in Figure 4. These sheets are connected along the [101] direction to form the three-dimensional framework by the bridging O atoms of the [LiO] layers. For the LiBaBO₃ crystal, however, the Ba atoms are coordinated by nine O atoms and the BaO₉ polyhedron is described as a mono-capped distorted square antiprism. The Ba–O distances vary from 2.622(6) to 3.185(5) Å with an average value of

2.813 Å, which compared well to the expected value 2.850 Å calculated from the crystal radii for the nine-coordinate Ba²⁺ ion.¹⁹ In the [BaO] layers, the Ba atoms through sharing the planes of three oxygen atoms extend along the *ac* diagonal direction to form chains, and the adjacent chains link together by the two oxygen atoms to form puckering sheets in the *b* direction, as shown in Figure 5. These sheets are connected along the [101] direction to form the three-dimensional framework by the bridging O atoms of the [LiO] layers.

3.2. Electronic Structures. The energy bands for LiSrBO₃ and LiBaBO₃ are calculated in terms of MO procedures. Note that the top of the valence band of LiSrBO₃ and LiBaBO₃ is at –7.07 and –5.65 eV, respectively, and is taken to be zero and regarded as a reference in the following discussion.

The dominant contribution to the lower valence band (i.e., –31.5 to –23.5 eV for LiSrBO₃, –28.5 to –24.5 eV for LiBaBO₃) comes from O 2s orbitals along with a relatively small (<8%) contribution from B 2s orbitals. It is therefore assigned as an s valence band. The upper

(19) Shannon, R. D. *Acta Crystallogr.* **1976**, *A32*, 751–767.

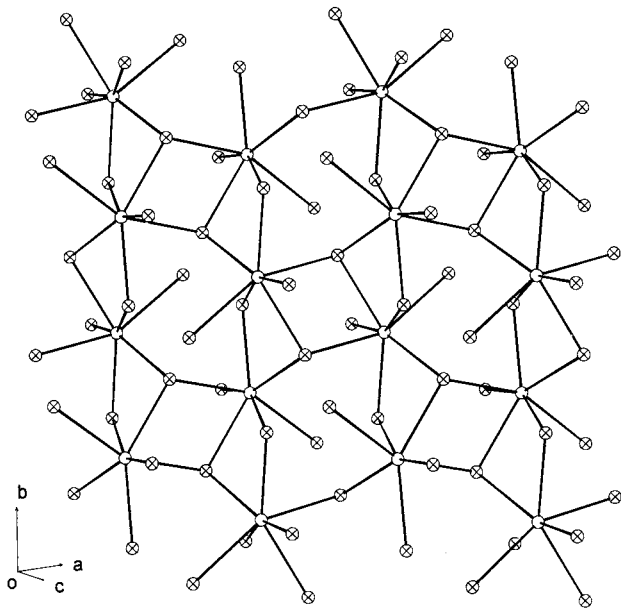


Figure 4. Structure of the [SrO] layer along the ac diagonal plane in $[\text{LiSrBO}_3]$.

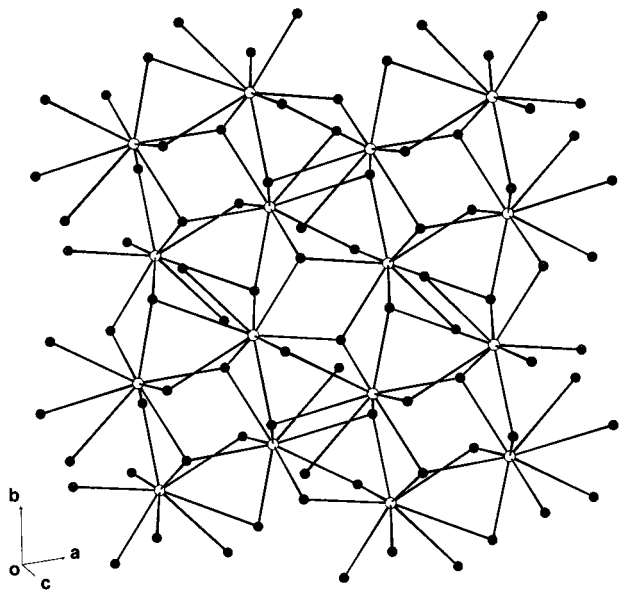


Figure 5. Structure of the [BaO] layer along the ac diagonal plane in $[\text{LiBaBO}_3]$.

valence band ranging from -18.8 to 0.00 eV is a result of contributions from O 2p orbitals that are mixed with B 2p orbitals (less than 37%) for LiSrBO_3 and that ranging from -9.5 to 0.00 eV is a result of contributions from O 2p orbitals that are mixed with B 2p orbitals (less than 45%) for LiBaBO_3 . The atomic state densities of the upper valence bands for LiSrBO_3 and LiBaBO_3 are displayed in Figure 6a,b, respectively. In Figure 6a, the σ -bonding interactions between the O 2p and B 2p orbitals yield the peaks ranging from -18.8 to -7.0 eV. On the other hand, the peaks localized in the energies from -5.0 to -2.3 eV can be assigned to the π -bonding interactions between the O 2p and B 2p orbitals. Finally, the 2p orbitals of the terminal oxygen atoms dominate the region from -1.9 eV to the top of the valence band. For example, the peak at the energy of -1.7 eV is the contribution from 89% O 2p orbitals. Figure 6a also includes the atomic state density of the lower conduction

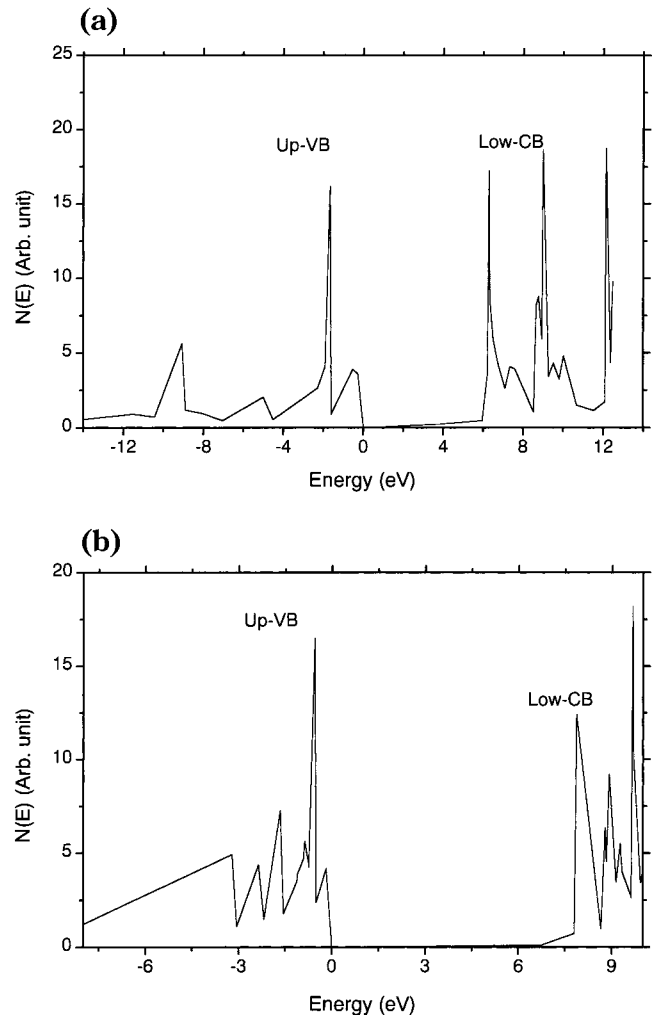


Figure 6. (a) Atomic state density of LiSrBO_3 . (b) Atomic state density of LiBaBO_3 .

band (CB) for LiSrBO_3 . Here, the peaks localized in the energy region between 3.9 and 13.5 eV are mainly due to contributions from the Sr and Li ionic states. For examples, the band of energy with less than 6.0 eV has contributions of about 58% from the Sr 5s orbital. The three peaks localized at 6.3 , 9.0 , and 12.1 eV have contributions of about 57% from Sr 4d orbitals and 17% from Sr 5p orbitals, 48% from Sr 4d and 38% from Li 2s orbitals, and 61% from Li 2p orbitals, individually. The conduction band region ranging from 14.1 to 25.0 eV is formed by the antibonding interactions between the orbitals of boron ($>60\%$ of 2s and 2p characters) and oxygen ($<40\%$ of 2p character). The calculated gap between the top of the valence band and bottom of the conduction band comes out to be 3.86 eV, yielding the absorption edge of the LiSrBO_3 crystal to be about 321 nm. In Figure 6b, the σ -bonding interactions between the O 2p and B 2p orbitals contribute in the lowest energy region of the p subband ranging from -9.5 to -3.2 eV. The band from the -3.1 to -1.5 eV is mainly due to the π -bonding interactions between the O 2p and B 2p orbitals. For example, the peak at the energy of -1.7 eV is the contribution from 88% O 2p and 4% B 2p orbitals. The band between the energies -1.1 and 0.0 eV is assigned to the 2p orbitals of the terminal oxygen atoms, and there are contributions of larger than 92% from O 2p orbitals. The three peaks localized at

about the energies of -0.9 , -0.6 , and -0.2 eV are the contributions of about 97, 95, and 93% from O 2p orbitals. The lower conduction band that ranges from 6.7 to 10.1 eV has contributions from the Ba and Li ionic states. For instance, the highest peak in Figure 6b has the contributions of about 59% from Ba 5d and 12% from Li 2p orbitals. The next band (10.7–15.8 eV) is formed by the antibonding interactions between the boron ($>30\%$ of 2s and 2p characters) and oxygen ($<46\%$ of 2s and 2p character) orbitals. The minimum energy gap between the top of the valence band and the bottom of the conduction band is calculated to be 6.70 eV, yielding the absorption edge of LiBaBO₃ to be about 185 nm.

3.3. Optical Properties. The optical properties of a bulk material can be thought of as being built up from the corresponding properties of individual cluster units or microscopic species. In this approximation, the linear refractive index of frequency dependence can be written as

$$n^2(\omega) = 1 + 4\pi\chi^{(1)}(\omega) \quad (2)$$

where the linear susceptibility $\chi^{(1)}(\omega)$ is defined as

$$\chi^{(1)}(\omega) = N\alpha(\omega)/[1 - (4\pi/3)N\alpha(\omega)] \quad (3)$$

Here, $\alpha(\omega)$ is the frequency-dependent polarizability and N is the cluster number density. For the usual case in which the polarizability $\alpha(\omega)$ is positive, we see that the susceptibility is larger than the value $N\alpha(\omega)$ predicted by calculations that ignore local-field corrections. The cluster number density, N , can be obtained by a product of mass density and Avogadro's number that is dividing by molar mass. For (LiSrBO₃)₂, with a molecular formula weight of 306.74 g/mol and a mass density of 1.82 g/cm³, we obtain $N = 3.573 \times 10^{21}$ cm⁻³. For (LiBaBO₃)₂, with a molecular formula weight of 406.18 g/mol and a mass density of 4.67 g/cm³, we obtain $N = 6.924 \times 10^{21}$ cm⁻³. It is to be noted here that, through rearrangement of the Lorentz–Lorenz equation, $(\epsilon^{(1)} - 1)/(\epsilon^{(1)} + 2) = 4/3\pi N\alpha$, we can obtain the relationship $(\epsilon^{(1)} + 2)/3 = 1/[1 - (4\pi/3)N\alpha]$.²⁰ Accordingly, the factor $1/[1 - (4\pi/3)N\alpha]$ or $(\epsilon^{(1)} + 2)/3$ is interpreted as the local-field correction factor for the linear susceptibility and can be described as the interaction between a selected cluster unit and surrounding environment. In fact, the determination of the local field factor can be quite complicated for a condensed system.^{21,22} We only take an approximate treatment of a spherical cavity to obtain the local field factor²³ in this study.

Before attempting to compute the frequency dependence of the refractive index for orthoborates LiSrBO₃ and LiBaBO₃, it is necessary to investigate the convergence behavior in the summation of excited states of clusters (LiSrBO₃)₂ and (LiBaBO₃)₂. This exercise will help us in determining the reliability of the INDO/SCI method used here. Figure 7 shows the calculated average polarizability $\langle\alpha\rangle$ versus the number of states for these two clusters at 1064 nm. Accordingly, $\langle\alpha\rangle$ is

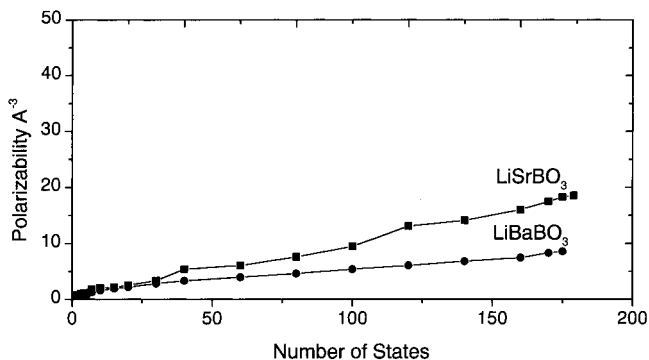


Figure 7. Polarizability $\langle\alpha\rangle$ clusters at 1065 nm.

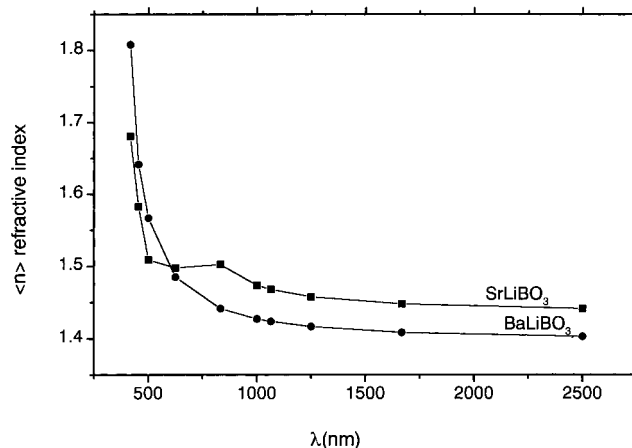


Figure 8. Average refractive index of SrLiBO₃ and BaLiBO₃ bulks.

converging very slowly, though it has reached the stability after summation over 150 states, and the convergent behavior is better for the (LiBaBO₃)₂ than for the (LiSrBO₃)₂. It is well-known that the sum-over-states expansion in eq 1 is, in general, infinite since the applied optical field mixes the ground state to several excited states. Combining the results of Figure 7 with the state density of Figure 6, we can identify that the charge transfers from the O 2p orbitals to metal ionic valence orbitals are the dominant contributors to the calculated linear polarizabilities.

The average dynamic refractive index of $\langle n(\omega) \rangle$ was calculated using a general definition for the average polarizability (i.e., $\langle a \rangle = (a_{xx} + a_{yy} + a_{zz})/3$) and eqs 2 and 3. Figure 8 shows the refractive indices $\langle n(\omega) \rangle$ of LiSrBO₃ and LiBaBO₃ for wavelengths ranging from 350 to 2500 nm. It is found that the dynamic $\langle n \rangle$ is larger for LiSrBO₃ than for LiBaBO₃ in the low-frequency zone. These are due to a smaller optical gap and larger dipole transition moment for LiSrBO₃ than for LiBaBO₃. For example, the calculated dipole transition moments of the first two low-lying excited states are 1.47 and 0.96 D for M = Sr and 0.95 and 0.18 D for M = Ba in the (LiMBO₃)₂ clusters. It also notes here that calculations of polarizabilities depend on the dipole transition moments, which in fact, depend on the state function and spatial configuration for LiMBO₃. At a wavelength of 1.065 μm, the calculated refractive indices, n_x , n_y , and n_z , are individually 1.574, 1.680, and 1.198, and the average $\langle n \rangle$ is 1.484 for SrLiBO₃; n_x , n_y , and n_z are individually 1.324, 1.571, and 1.388, and the average $\langle n \rangle$ is 1.428 for LiBaBO₃. The refractive indices of these

(20) Boyd, R. W. *Nonlinear Optics*; Academic Press: San Diego, CA, 1992; p 148.

(21) Munn, R. *Int. J. Quantum Chem.* **1992**, *43*, 159–169.

(22) Sylvester-Hvid, K. D.; Mikkelsen, K. V.; Patrick, D. J.; Agren, H. J. *J. Phys. Chem. A* **1999**, *103*, 8375–8383.

(23) Burland, D. M.; Miller, R. D.; Walsh C. A. *Chem. Rev.* **1994**, *94*, 31–75.

two crystals have not been observed, and our calculated values are only an estimation. Generally, the observed average refractive index of borate compounds is from about 1.50 to 1.70 in a low-frequency zone.²⁴ Given the approximations involved in the computational model used here, the calculated average refractive indices of the two orthoborates seem to be underestimations of about 10%.

4. Conclusions

In this work, the new orthoborates are reported in single-crystal structural determinations and synthetic procedures, and their cluster units $(\text{LiMBO}_3)_2$ are chosen to calculate the electronic structures and refractive indices of LiMBO_3 ($M = \text{Sr}, \text{Ba}$). Electronic structure calculations find the lower valence band derived from O 2s and B 2s orbitals whereas the upper valence

band was formed by the bonding interactions between O 2p and B 2p orbitals; electronic structure calculations find the lower conduction band derived from metal ionic orbitals whereas the upper conduction band was formed by the antibonding interactions between O and B valence orbitals. The closer to the top of the valence band, the more contributions from the O 2p orbitals; the closer to the top of the conduction band, the more contributions from the B valence orbitals. The charge transfer from O 2p orbitals to cationic orbitals is found to make significant contributions to the calculated refractive index of these compounds, and the refractive index is larger for LiSrBO_3 than for LiBaBO_3 .

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(24) Davis, H. M.; Knight, M. A. *J. Am. Ceram. Soc.* **1945**, *28*, 97–102.